



Low-Pt supported on MOF-derived Ni(OH)₂ with highly-efficiently electrocatalytic seawater splitting at high current density

Jianpeng Sun^a, Zisheng Zhang^b, Xiangchao Meng^{a,*}

^a Key Laboratory of Marine Chemistry Theory and Technology (Ministry of Education), College of Chemistry & Chemical Engineering, Ocean University of China, Qingdao, Shandong 266100, China

^b Department of Chemical and Biological Engineering, Faculty of Engineering, University of Ottawa, Ottawa, Ontario K1N6N5, Canada



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ABSTRACT

Developing highly effective and stable electrocatalysts are attractive but highly challenging for alkaline seawater splitting. Herein, using a simple method, we developed a novel MOF derived nanosheets-nanoparticle-structured Pt₂/Ni(OH)₂/NF catalyst. Benefiting from strong metal-support interaction (SMSI) of Pt-O-Ni bond at interface, the synergistic effect of Pt-O vacancies (Ov) sites and excellent superhydrophilic surface, catalyst delivered excellent catalytic activity in seawater splitting. The overpotential was 19 mV at 10 mA·cm⁻², which surpassed the commercial 20% Pt/C. A photovoltaic cell yielded up to 13.5% STH conversion efficiency. Furthermore, the cell voltage at 10 mA·cm⁻² for anion exchange membrane (AEM) seawater electrolyzer in our design with the Pt₂/Ni(OH)₂/NF cathode catalyst was merely 1.46 V. The energy consumption to produce 1 m³ of H₂ at the current density of 200 mA·cm⁻² was 3.8 kW·h, below that of NF= |NF (4.3 kW·h). Theoretical results showed that the localized charge density and electronic structure of Ni(OH)₂ can be significantly modulated upon SMSI at interface and Ov. And the synergistic effect of Pt-Ov sites can balance the adsorption and desorption of H^{*}. Ov could achieve high-efficiency adsorption of H₂O and optimize hydrogen spillover from Ni(OH)₂ to Pt, leading to excellent HER activity. This study provided a light in rational construction of robust seawater-based electrocatalysts for seawater splitting.

1. Introduction

Seawater, as an alternative water resource for electrocatalytic hydrogen evolution has increasingly attracted attentions [1–3]. However, the complexity of seawater resulted in the low activity and instability of electrocatalysts when applied in seawater splitting [4–6]. Compared with fresh water, the large content of Cl⁻ in seawater may block the active sites, preventing adsorption of H^{*} and then leading to slow HER kinetics [7,8]. Moreover, Cl⁻ may cause corrosion and gradual dissolution of electrocatalyst, leading to catalyst poisoning and poor catalytic stability [9,10]. Therefore, it is urgent to design and prepare a highly effective and stable electrocatalyst for seawater splitting to evolve H₂.

Layered double hydroxides (LDHs) such as Ni(OH)₂ exhibited promises in electrocatalytic water splitting [11–13]. Nevertheless, strong H^{*} adsorption capacity on its surface prevented the release of hydrogen, which further limited the electrocatalytic activity [14]. Pt has been designed in highly-efficiently electrocatalytic hydrogen evolution

[15,16]. Due to the weak interaction between 4d orbital of Pt and 2p orbital of oxygen in H₂O, the adsorption and dissociation of H₂O on Pt was poor, especially in alkaline aqueous solution [17,18]. Considering the sluggish HER kinetics in alkaline seawater, designing and engineering seawater-based nanomaterials interfaces by coupling low-Pt with Ni(OH)₂ is considered to be an effective method. Zhao et al. designed Pt decorated Ni–Ni(OH)₂ to enhance reaction kinetics, which delivered high electrocatalytic activity [19]. Zhang et al. also reported a robust Pt/TiO₂/Ni(OH)₂ nanosheet arrays, which showed good performance for high current density alkaline water electrolysis due to the synergistic effect of Pt and TiO₂/Ni(OH)₂ [20]. Although many interface engineering research has been used to design highly reactive electrocatalyst, few reports on coupling Pt with Ni(OH)₂ to construct an high stability seawater-based catalyst, which was mainly because the complex ionic environment is easy to destroy the active site of the in seawater. Recently, strong metal-support interaction (SMSI) in catalysts, as an effective behavior to enhance the stability, has attracted extensive attentions[21]. Especially, the construction of M-O-M (M = metal)

* Corresponding author.

E-mail address: mengxiangchao@ouc.edu.cn (X. Meng).

bonds at interface can protect active sites well, which made electrocatalysts keep good stability [22,23]. At present, metal–organic frameworks (MOFs) as templates have been widely studied for transition metal-based electrocatalysts due to their tunable nanostructures, and excellent porosities [24,25]. Using MOFs as templates or precursors of Ni(OH)_2 can inherit the inherent structural advantages of MOFs [26]. Meanwhile, the surface of MOF is easy to generate O vacancies (Ov) and form M–O–M bonds during the in-situ conversion process, which can effectively prevent its active site aggregation and then improving the stability of Ni(OH)_2 in alkaline seawater splitting [27–29]. Therefore, approaches to couple Pt and SMSI effect on MOF derived Ni(OH)_2 would be an useful strategy to further improve electrocatalytic activity and stability in alkaline seawater splitting.

Based on above discussions, we developed a novel MOF derived nanosheets-nanoparticle-structured Pt2/ Ni(OH)_2 /NF catalyst. SMSI of Pt–O–Ni bond at interface, synergistic effect of Pt–O vacancies (Ov) sites and excellent superhydrophilic surface in Pt2/ Ni(OH)_2 /NF promoted the electrocatalytic activity. Furthermore, cell voltage at 10 $\text{mA}\cdot\text{cm}^{-2}$

for the anion exchange membrane (AEM) seawater electrolyzer was only 1.46 V. The energy consumption to produce 1 m^3 of H_2 at the current density of 200 $\text{mA}\cdot\text{cm}^{-2}$ was 3.8 kW·h, below that of NF=|NF (4.3 kW·h). The mechanism of as-prepared sample in alkaline seawater splitting was explored based on both experimental and theoretical computation results.

2. Experimental section

2.1. Preparation of Ni-MOF/NF

In a typical synthesis process, $\text{NiCl}_2\cdot 6 \text{ H}_2\text{O}$ (44 mg) and terephthalic acid (62 mg) were separately dispersed in N,N-Dimethylformamide (DMF, 16 mL), ethanol (1 mL) and deionized (DI) water (1 mL) under ultrasonication. And the mixture was transferred into an autoclave, and a piece of NF (2 cm × 2 cm) was vertically immersed into the solution, which was kept at 120 °C for 12 h. As-synthesized product was obtained at 70 °C under vacuum for 12 h.

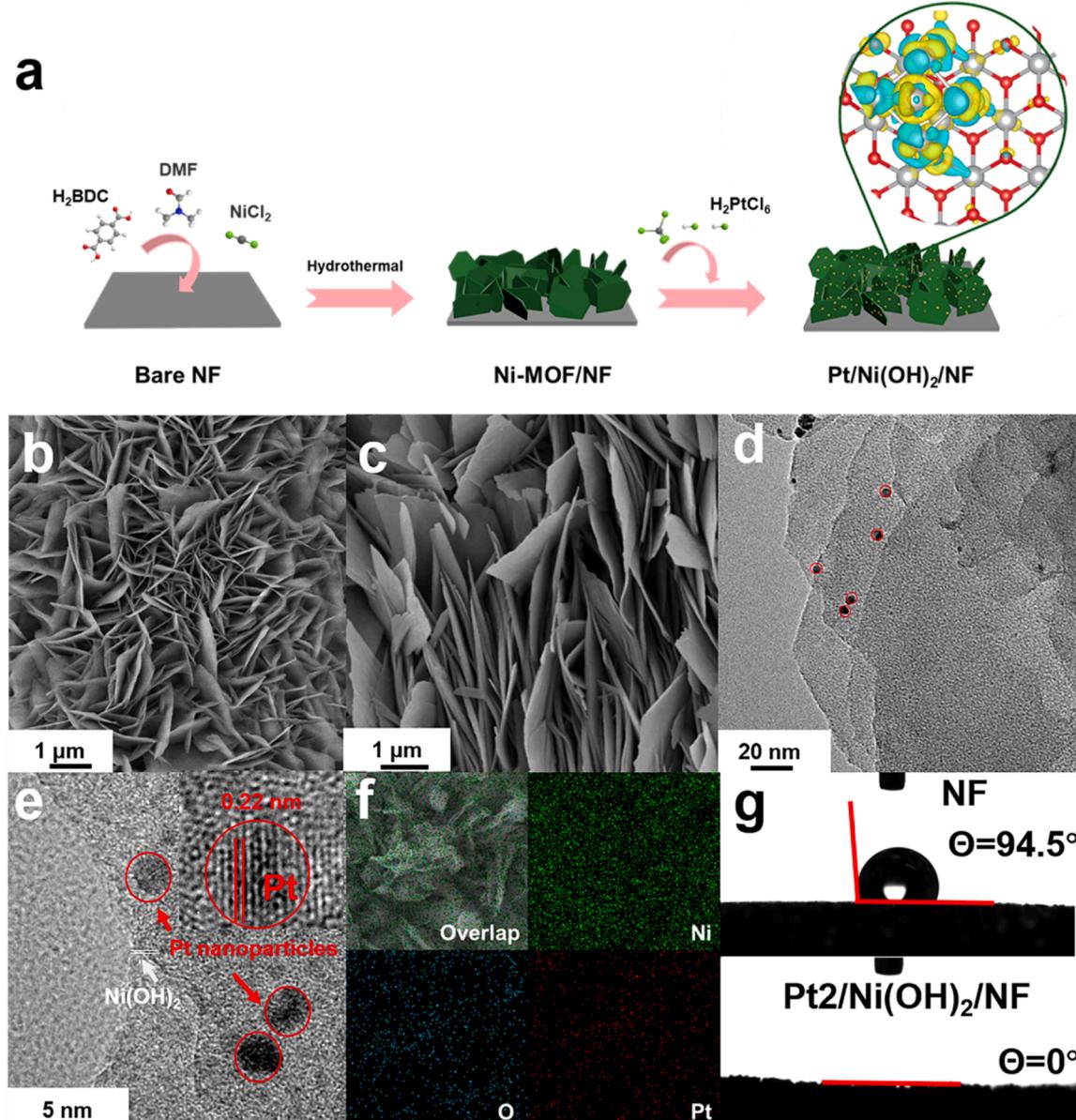


Fig. 1. (a) Schematic illustration of the synthesis of Pt2/ Ni(OH)_2 /NF. SEM images of (b) Ni-MOF/NF and (c) Pt2/ Ni(OH)_2 /NF. (d) TEM image, (e) HRTEM image and (f) EDS mapping images of Pt2/ Ni(OH)_2 /NF. (g) The droplet contact angle images of the bare NF and Pt2/ Ni(OH)_2 /NF.

2.2. Preparation of Pt/Ni(OH)₂/NF

A hydrothermal method was applied to prepare Pt/Ni(OH)₂/NF. Typically, H₂PtCl₆ (10 mg/mL: 240 μ L, 480 μ L and 720 μ L) was dissolved in a Teflon-lined autoclave containing 15-mL DI water. And then a piece of Ni-MOF/NF (2 cm \times 2 cm) was vertically immersed into the solution. Afterwards, the autoclave was kept at 150 °C for 460 min. Finally, Pt/Ni(OH)₂/NF was collected after washing.

3. Results and discussions

3.1. Synthesis and characterizations

Fig. 1a showed Ni-MOF-74 was chosen as a template to form Ni(OH)₂ due to their similar crystal structure (**Fig. S1**), and grown on NF using a hydrothermal method [30]. As-prepared Ni-MOF-74 was further treated to prepare Ni(OH)₂ using a alkaline hydrolysis at 150 °C (**Fig. S1**). From SEM and XRD tests (**Fig. 1b** and S2), Ni(OH)₂ was successfully constructed in NF. And then, Ni(OH)₂ was modified with three different amount of H₂PtCl₆. The content of Pt was calculated as 0.56%, 1.12% and 1.68 wt% in the composite, which was labelled as Pt1/Ni(OH)₂/NF, Pt2/Ni(OH)₂/NF and Pt3/Ni(OH)₂/NF, respectively. Pt2/Ni(OH)₂/NF maintained nanosheets morphology (**Fig. 1c**). TEM has been applied to investigate Pt in the composite in **Figs. 1d** and **1e**. Pt nanoparticle had a fringe spacing of 0.22 nm, which was in consistent with the Pt [1 1 1] planes. And particles with a diameter of ~4 nm were found on nanosheets because of the SMSI of Pt-O-Ni bond at interface after in-situ MOF

conversion process. From XRD patterns of Pt2/Ni(OH)₂/NF, peaks for Pt were also found, which were indexed in **Fig. S3**. In addition, the TEM image of pure Ni(OH)₂/NF was also tested in **Fig. S4**, and no particles were observed. Simultaneously, EDS mapping delivered the good distribution of O, Pt, Ni species (**Fig. 1f** and S5). Previous studies have been proved that the electrolyte ions were insufficiently adsorbed on the electrode surface, and the catalytic active sites could not be exposed to the maximum extent, resulting in poor electrocatalytic activity [31,32]. The construction of hydrophilic structure had a good effect on the catalytic reaction. And then, hydrophilicity of bare NF and Pt2/Ni(OH)₂/NF was investigated in **Fig. 1g**. Pt2/Ni(OH)₂/NF was superhydrophilic and had a contact angle closed to 0°, it was much smaller than that of NF substrate (94.5°), indicating that sites in Pt2/Ni(OH)₂/NF had good contact with electrolyte. Overall, Pt2/Ni(OH)₂/NF had good hydrophilicity and large specific surface area, which would accelerate the adsorption of electrolyte, and then achieving good electrocatalytic activity.

The compositions of as-prepared samples were investigated using XPS. The survey spectrum was depicted in **Fig. S6**, peaks for Ni, O, Pt and C were indexed, and no impre peaks were observed. The typical peaks of Ni³⁺ and Ni²⁺ were shown in **Fig. 2a** [33]. Compared with that of Ni(OH)₂/NF, peaks of Pt2/Ni(OH)₂/NF had a negatively shifts about ~0.41 eV, which meant electrons were accumulated at Ni species because of the formation of the strong metal-support interaction of Pt-O-Ni bond at interface after in-situ MOF conversion process in Pt2/Ni(OH)₂/NF [22]. The concentration of electrons could optimize the adsorption capacity of H⁺. And Pt-O-Ni bond at interface could protect

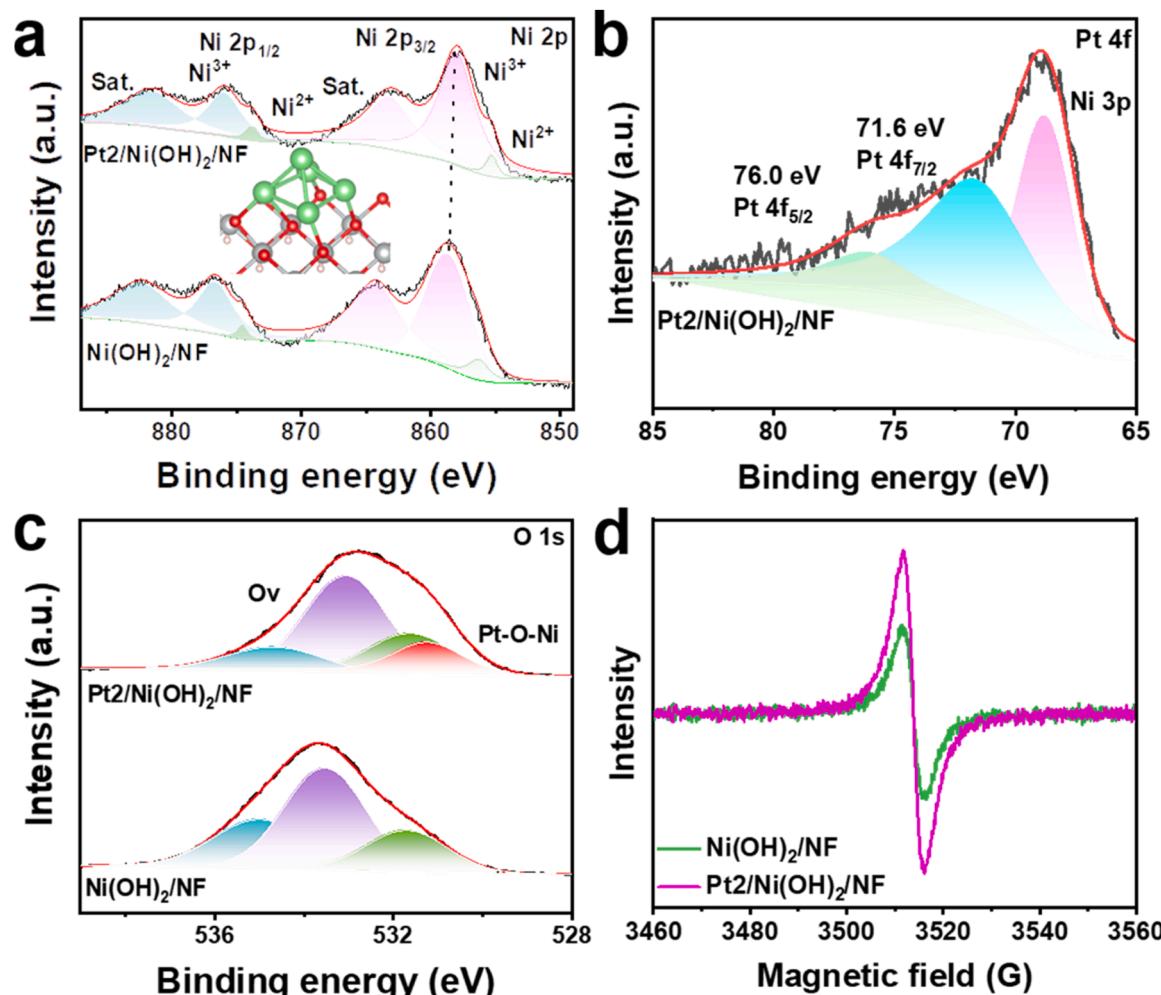


Fig. 2. XPS spectra of (a) Ni 2p, (b) Pt 4 f, (c) O 1 s (d) EPR spectra.

the active site during the electrocatalytic tests and make Pt₂/Ni(OH)₂/NF maintain high reusability. Moreover, two characteristic peaks were indexed to Pt 4 f_{7/2} and Pt 4 f_{5/2} of Pt²⁺ species in Fig. 2b, which also implied the presence of a Pt–O–Ni structure [34]. SMSI between nanoparticles and Ni(OH)₂ could ensure its stable existence of Pt nanoparticles in HER process. Moreover, induced by the SMSI of Pt–O–Ni bond at interface, binding energy of lattice oxygen of Pt₂/Ni(OH)₂/NF shifted to a lower level (531.6 eV) different from Ni(OH)₂/NF (531.7 eV) in Fig. 2c. And the peaks at 531.23 eV were ascribed to the Pt–O–Ni bond [34]. Meantime, the content of Ov in Pt₂/Ni(OH)₂/NF also increased significantly from 50.47% to 54.99%. EPR spectra were shown to further verify the presence of Ov on Pt₂/Ni(OH)₂/NF. Ni-MOF-74 readily formed Ov on Ni(OH)₂ at high temperatures (Fig. 2d). More importantly, more Ov in Pt₂/Ni(OH)₂/NF were detected, which might be because the loaded Pt reduces the generation energy of Ov [35]. Overall, after forming Pt–O–Ni structure and Ov, electron structure has changed in Pt₂/Ni(OH)₂/NF, which could enhance strong metal-support interaction and ensure its stable existence of active sites in alkaline seawater splitting.

3.2. Electrochemical HER performance

HER activities of Pt₂/Ni(OH)₂/NF and other contrast samples were

studied in alkaline seawater solution. Comparatively, Pt₂/Ni(OH)₂/NF delivered a high activity (19 mV), which was pretty lower than bare NF (241 mV) and Ni(OH)₂/NF (186 mV) at 10 mA·cm⁻² and even better than 20% Pt/C (21 mV) (Fig. 3a). Meantime, Pt₂/Ni(OH)₂/NF only required HER overpotentials of 76 and 283 mV at 100 and 1000 mA cm⁻², respectively. As displayed in Fig. 3b, the Tafel value went from 189 mV·dec⁻¹ for Ni(OH)₂/NF and 281 mV·dec⁻¹ for NF to 70 mV·dec⁻¹ for Pt₂/Ni(OH)₂/NF after the Pt load, indicating that HER kinetics was enhanced for Ni(OH)₂ with Pt modification [36]. Moreover, Pt₂/Ni(OH)₂/NF still showed excellent activity, which only needed 274 mV at 1000 mA cm⁻² and had small Tafel value in 1.0 M KOH (Fig. S7). Such activity could be because: (i) Pt nanoparticles and Ov could effectively lower the adsorption of H^{*} and then achieve high-efficiency adsorption of H₂O; (ii) The strong metal-support interaction in Pt–O–Ni bonds could optimize the electronic structure and achieve fast electronic transfer; (iii) Superhydrophilic surface could ensure rapid release of gases at a high current density, and then improving HER activity. Actually, the activity of Pt₂/Ni(OH)₂/NF in 1.0 M KOH was better than the activity in alkaline seawater, which may be caused by due to the critical issue of Cl⁻ contamination on the Pt surface. Therefore, the cyclic voltammetry at 0.7–0.8 V were tested in Fig. S8, Pt₂/Ni(OH)₂/NF had a larger ECSA in 1.0 M KOH than that in alkaline seawater [37]. This meant that Cl⁻ contamination on the Pt

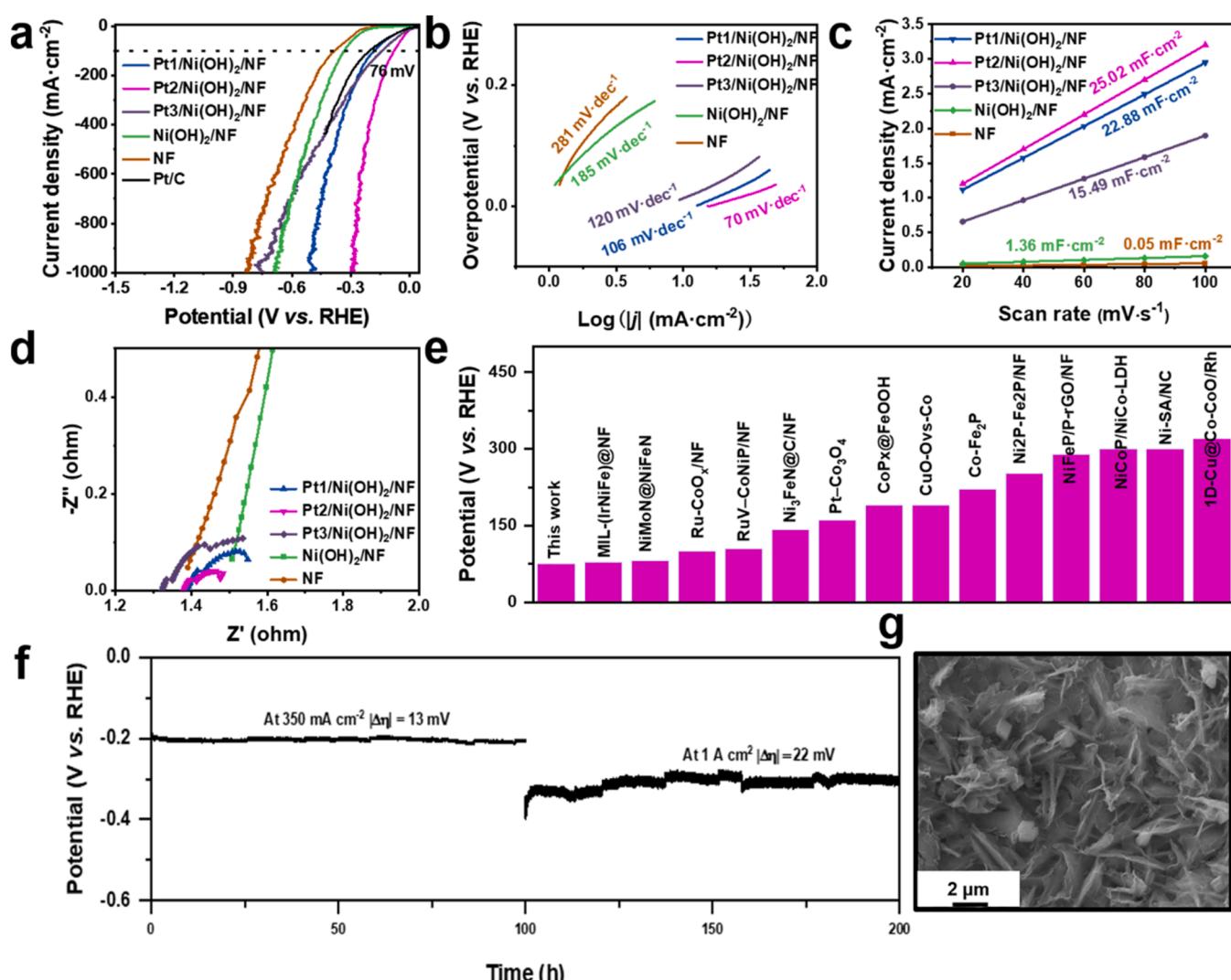


Fig. 3. (a) LSV curves. (b) Tafel plots. (c) Electrochemical double layer capacitance (C_{dl}). (d) EIS. (e) Performance comparison diagram. (f) Long-term stability measurements. (g) SEM image after stability tests.

surface in alkaline splitting could lower the active area and then affect the activity of Pt₂/Ni(OH)₂/NF. The double-layer capacitance (C_{dl}) for Pt₂/Ni(OH)₂/NF was 25.02 mF·cm⁻² in Fig. 3c. This meant that the catalyst had a high ECSA to expose more reactive sites. In addition, the Nyquist diagram (Fig. 3d) of Pt₂/Ni(OH)₂/NF was the smallest among samples, which further indicated that faster charge transfer and promoted hydrogen evolution kinetics after introduction of Pt nanoparticles and Ov. More importantly, compared to other recently reported seawater-based electrocatalysts in Fig. 3e, Pt₂/Ni(OH)₂/NF also

delivered excellent HER property from the aspect of overpotential at 100 mA cm⁻². Furthermore, catalytic stability was carried out by amperometric stability for 200 h (Fig. 3f). Actually, there was negligible changes due to the strong Pt-O-Ni interaction. And SEM and XRD images (Fig. 3g and S9) of Pt₂/Ni(OH)₂/NF after stability tests also showed its morphological features and phase structure, it could be seen that Pt₂/Ni(OH)₂/NF was transformed into rough due to the long time surface reactions in alkaline seawater. Such result meant Pt₂/Ni(OH)₂/NF had high structural and catalytic robustness and application prospect in

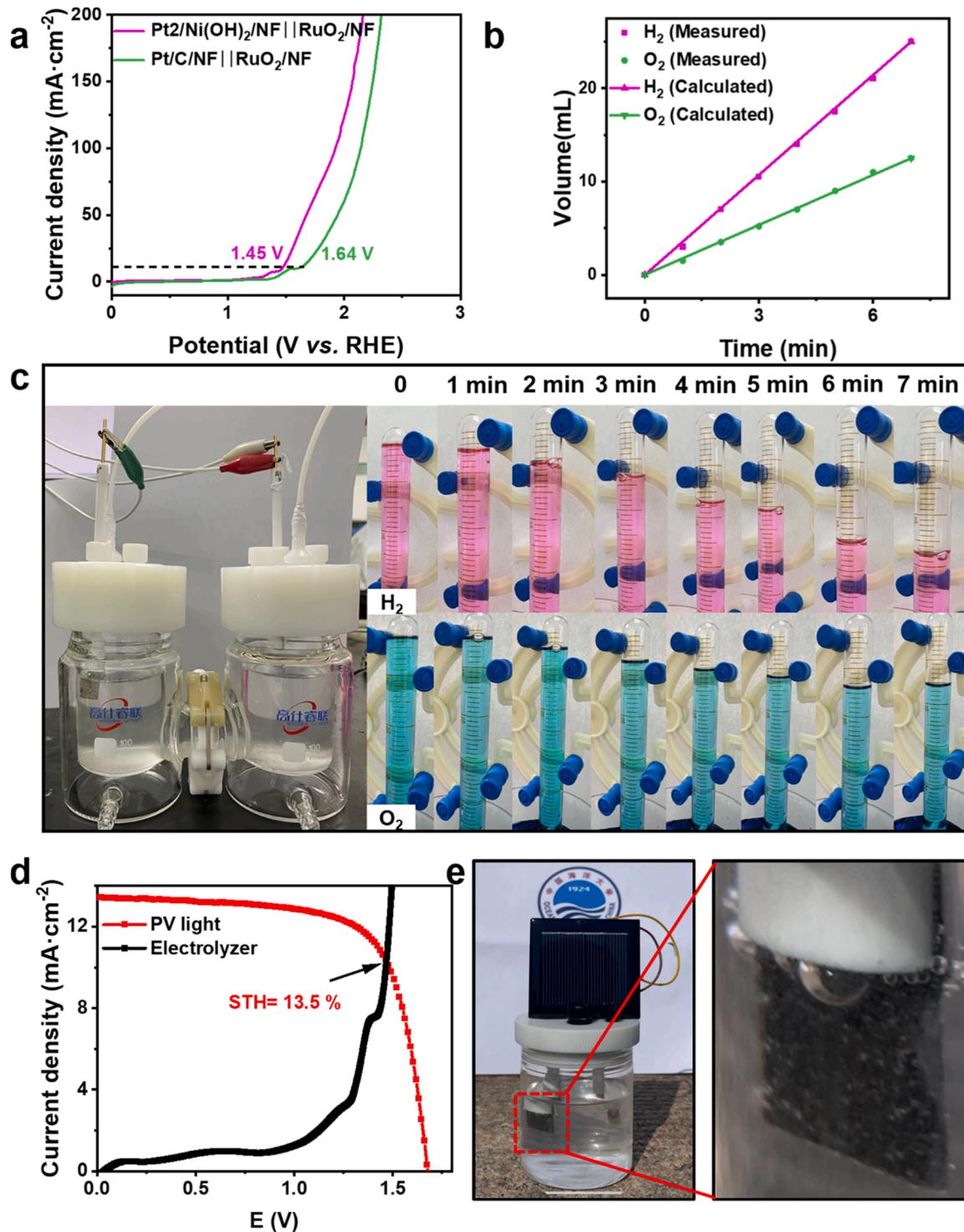


Fig. 4. (a) LSV curves. (b) Device diagram for measuring Faraday efficiency. (c) Enlargement of a drainage device in seawater media. (d) Current density–potential curves of the seawater electrolyzer and PSC. (e) Typical picture exhibiting the generation of H₂ gas during overall seawater splitting under natural light.

seawater splitting due to the SMSI effect at interface.

3.3. Overall seawater-splitting measurement

A seawater splitting cell was constructed by using Pt2/Ni(OH)₂/NF as cathode material and bare RuO₂/NF as anode material in Fig. S10. Compared with Pt/C/NF=|RuO₂/NF (1.64 V) in alkaline seawater, Pt2/Ni(OH)₂/NF electrolyzer merely required 1.45 V at 10 mA cm⁻² (Fig. 4a). Additionally, the Faraday efficiency (FE) was tested using the drainage method in Fig. 4b and H-type electrolytic cell was used to collect H₂ and O₂ in Fig. 4c. Fig. 4b proved that the FE was nearly 100%, indicating that catalysts had high selectivity in electrocatalytic seawater splitting [38]. Furthermore, Fig. 4d showed the photo-driven water-splitting system under simulated AM 1.5 G (intensity: 100 mW cm⁻²) illumination. In order to obtain sufficient photovoltage for seawater electrolysis, two perovskite solar cells (PSC) were connected in series and connected with Pt2/Ni(OH)₂/NF electrolytic cells to form a solar-driven electrolytic system. Fig. 4d showed a solar to hydrogen (STH) efficiency as high as 13.5% in alkaline seawater splitting [39]. This result was better than that of coupled normal Si solar cell (Fig. S11), and even higher than most of other reported works in Table 1. By coupling solar cell with the electrolyzer, many bubbles could be seen under natural light (actual voltage was 1.71 V) in Fig. 4e and S12 [26, 40]. Based on electrocatalytic testing results, Pt2/Ni(OH)₂/NF revealed excellent catalytic performance from the aspects of activity and the stability in alkaline seawater splitting for hydrogen production, which were capable for AEM electrolyzer.

3.4. AEM electrolyzer measurement

To realize the high-efficiency hydrogen production in industry, AEM

electrolyzer were assembled. Pt2/Ni(OH)₂/NF as cathode material in AEM water electrolyzer were used to verify their performance in practical devices [41–43]. The components inside the AEM water electrolyzer were shown in Fig. 5a. Here, alkaline exchange membranes were used in our device. As nickel-based materials were stable and mostly used as anode and cathode catalysts in industry. Bare NF materials were used as electrodes for comparison. AEM water electrolyzer had a significant H₂/O₂ bubble formation as shown in Fig. 5b. At the flow rate of the electrolyte of 30 mL·min⁻¹, AEM electrolyzer with Pt2/Ni(OH)₂/NF cathode catalyst was only 1.46 V at 10 mA cm⁻² and 1.60 V for the unit with the bare NF (Fig. 5c). Hydrogen production rate was calculated as 1425 L·h⁻¹, and the energy consumption to produce 1 m³ of H₂ was 3.8 kW·h, which was lower than NF=|NF (4.3 kW·h) at the current density of 200 mA·cm⁻² (Fig. 5d). The above results further confirmed the excellent activity of as-prepared Pt2/Ni(OH)₂/NF in electrocatalytic applications in practice.

3.5. Density functional theory (DFT) calculations

DFT calculations were applied to investigate the intrinsic electronic structure to study intrinsic activity of Pt2/Ni(OH)₂/NF. Localized charge distribution in Ov-Pt/Ni(OH)₂ model was simulated to study the strong metal-support interaction of Pt-O-Ni bonds of Pt2/Ni(OH)₂/NF [44]. Obviously, affected by the SMSI effect of Pt-O-Ni bonds, the localized charge distribution (Fig. 6a and 6b) of Pt/Ni(OH)₂ and Ov-Pt/Ni(OH)₂ underwent a symmetry breaking charge transfer in Ni(OH)₂ at the interface. This would result in a non-uniform charge distribution and electron redistribution [45]. Notably, unlike Pt/Ni(OH)₂, the generation of Ov led to the aggregation of electrons on adjacent Ni atoms in red circle, which optimized its H^{*} adsorption ability. And electron aggregation on Pt nanoparticles due to SMSI effect of Pt-O-Ni bonds could

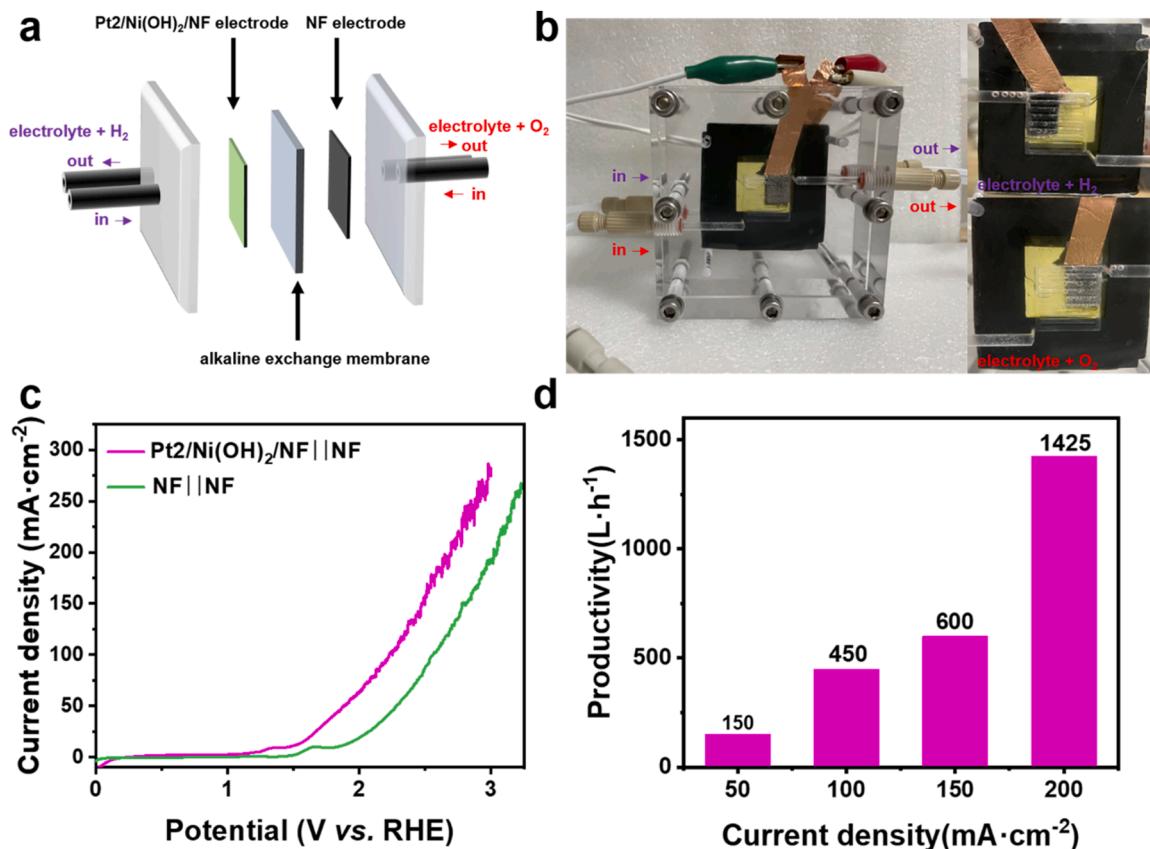


Fig. 5. (a) Schematic picture of AEM water electrolyzer. (b) Typical picture of AEM water electrolyzer. (c) LSV curves of the electrolyzer (without iR compensation). (d) Plot of calculated hydrogen production of Pt2/Ni(OH)₂/NF.

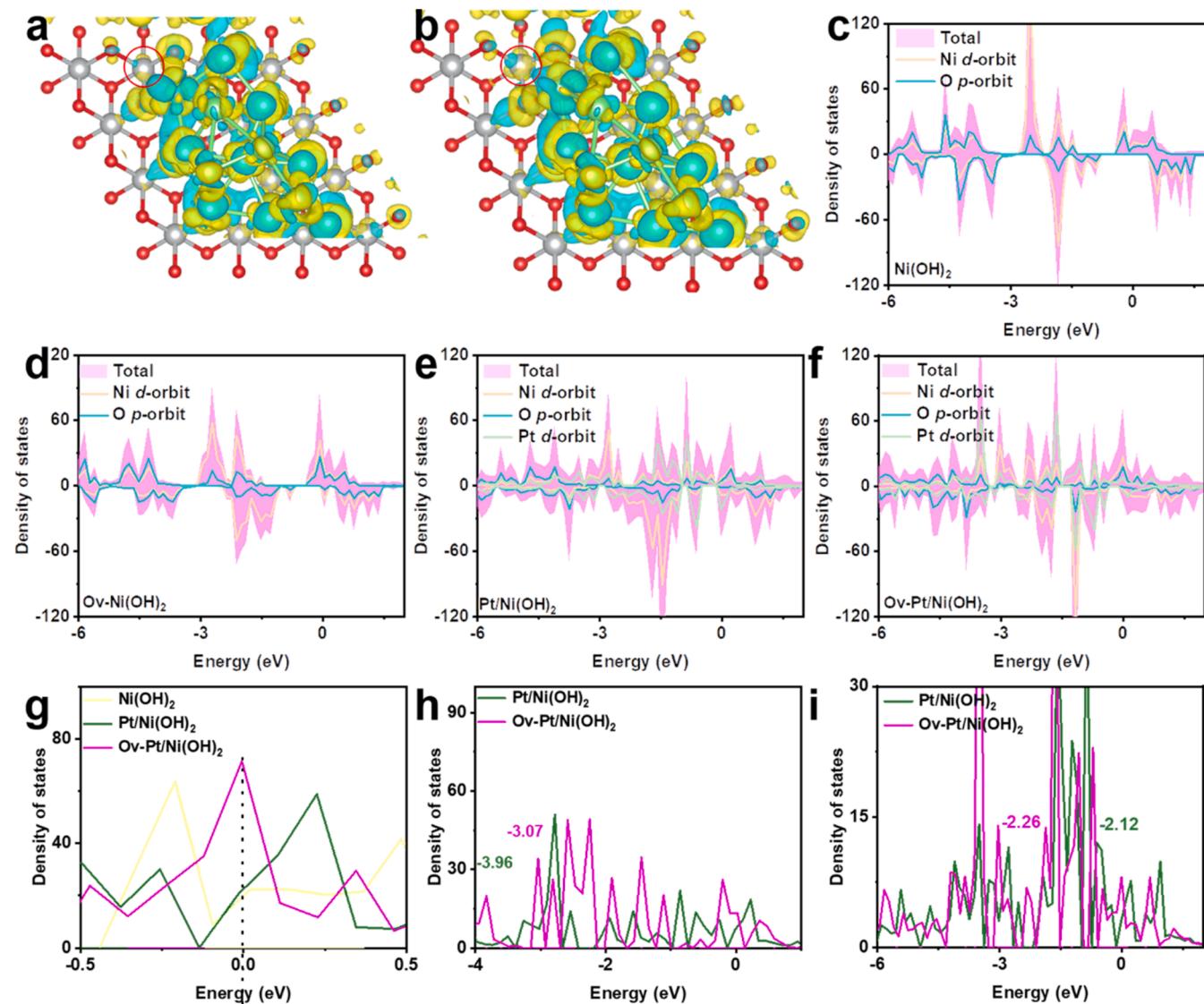


Fig. 6. Atomic models with charge density difference plot of (a) Pt/Ni(OH)₂ and (b) Ov-Pt/Ni(OH)₂. DOS of (c) Ni(OH)₂, (d) Ov-Ni(OH)₂, (e) Pt/Ni(OH)₂ and (f) Ov-Pt/Ni(OH)₂. (g) TDOS. D-band centers of (h) Ni atoms and (i) Pt atoms.

promote the adsorption of H^{*}. The schematic models were calculated in Fig. S13. As displayed in Figs. 6c - 6f, it could be seen that SMSI effect of Pt-O-Ni bonds, and Ov could effectively change the electronic structure of Ni(OH)₂ due to its metallic character with zero bandgap. Of note, after forming SMSI and Ov, total DOS (TDOS) of Ov-Pt/Ni(OH)₂ at the Fermi level was higher than that of Pt/Ni(OH)₂ and Ni(OH)₂ (Fig. 6g). Such result suggested strong interaction in Pt-O-Ni could efficiently achieve the fast electrons transfer rate and enhance electrical conductivity [46].

Moreover, d-band center was used to reflect the electronic interactions between the adsorbate and metal [47,48]. Compared with that of Pt/Ni(OH)₂ (-3.96 eV), d-band center of Ni was -3.07 eV (Ov-Pt/Ni(OH)₂) in Fig. 6h. Notably, the formation of Pt nanoparticles could increase the d-band central energy level and lead to the strong Ni-H interaction, improving electrocatalytic activity. Fig. 6i also proved that the presence of Ov could weaken the H^{*} adsorption on Pt sites. In short, the strong interaction in Ov-Pt/Ni(OH)₂ can effectively regulate H^{*} adsorption and desorption on active sites, and then and optimize its intrinsic HER performance [49].

The ΔG_{H^*} was applied for in-depth prediction of the synergy of Pt nanoparticles and Ov. Of note, when ΔG_{H^*} was close to 0, it meant that catalyst had a superior H^{*} adsorption capacity [50]. From Fig. 7 and S14, the ΔG_{H^*} value (-0.50 eV) at the Pt2 site in Pt/Ni(OH)₂ was better

than that at the Pt1 site (-0.63 eV) and O site (-0.69 eV), suggested that the adsorption activity of the H^{*} intermediate in Pt/Ni(OH)₂ was stronger at the Pt2 site than other sites. Meantime, the introduction of Pt nanoparticles greatly improved ΔG_{H^*} (-1.07 eV) of Ni(OH)₂. Furthermore, the adsorption energy had a minimum of -0.20 eV in all samples after the formation of Ov in Pt/Ni(OH)₂, which indicated that the synergistic effect of Pt-Ov sites could effectively regulate H^{*} adsorption and promote H₂ release.

Recently, hydrogen spillover as a novel method to regulate the activity had been widely studied [51,52]. When the ΔG_{H^*} value of active site (M) was more negative, the M-H adsorption was stronger. Otherwise, the M-H adsorption was weaker. Therefore, when H^{*} adsorption from the active site was stronger to weaker, it meant that H₂ was easy to spill out. The free energy values of HER on the Pt sites and O sites through the hydrogen-spillover pathway was summarized in Fig. 8a. Firstly, It could be seen that $\Delta G_{\text{H}_2\text{O}}$ on the Ov-Pt/Ni(OH)₂ surface (-0.51 eV) was obviously lower than that on the Pt/Ni(OH)₂ surface (-0.49 eV), suggesting Ov could achieve high-efficiency adsorption of H₂O, which was beneficial to promote above Volmer process ($\text{H}_2\text{O} + e^- = \text{H}^* + \text{OH}^-$). And then, ΔG_{H^*} value (-0.50 eV) of Pt2 sites in Pt/Ni(OH)₂ was closer to zero than other sites, which implied Pt2 sites were most active sites. If H could be transferred to Pt nanoparticles across the

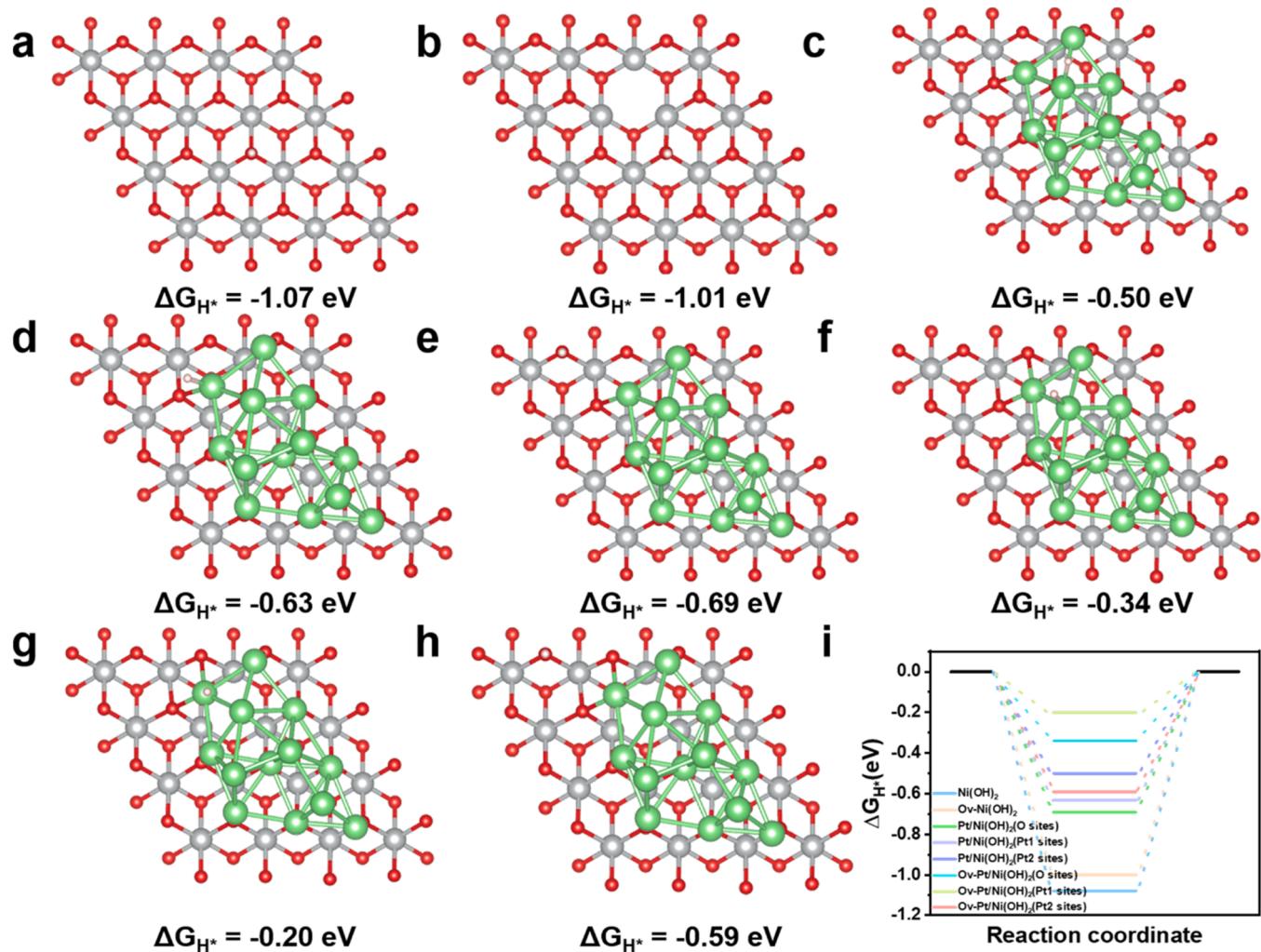


Fig. 7. The top-view schematic models of (a) $\text{Ni}(\text{OH})_2\text{-O}$ sites, (b) Ov- $\text{Ni}(\text{OH})_2\text{-O}$ sites, (c) Pt/ $\text{Ni}(\text{OH})_2$ -Pt2 sites, (d) Pt/ $\text{Ni}(\text{OH})_2$ -Pt1 sites, (e) Pt/ $\text{Ni}(\text{OH})_2\text{-O}$ sites, (f) Ov-Pt/ $\text{Ni}(\text{OH})_2$ -Pt2 sites, (g) Ov-Pt/ $\text{Ni}(\text{OH})_2$ -Pt1 sites and (h) Ov- $\text{Ni}(\text{OH})_2\text{-O}$ sites. (i) ΔG_{H^*} .

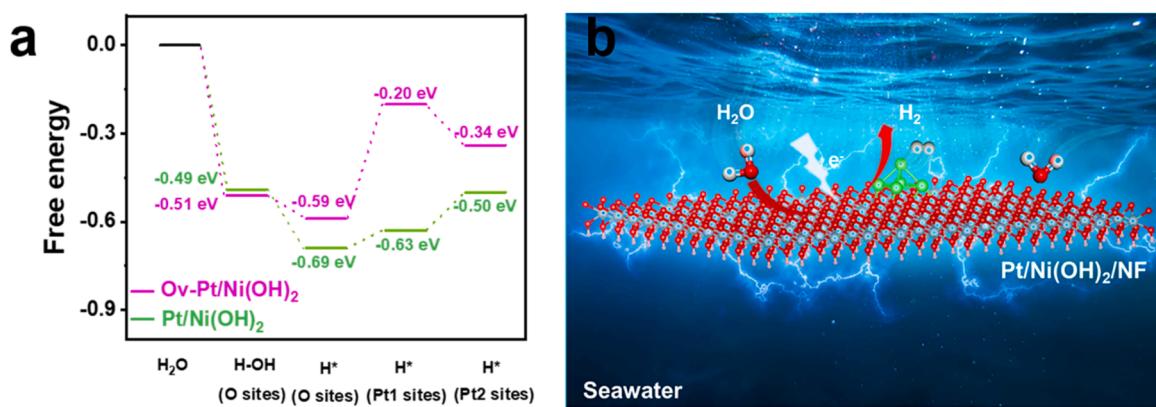


Fig. 8. (a) Calculated adsorption free energy diagrams onto the Pt/ $\text{Ni}(\text{OH})_2$ and Ov-Pt/ $\text{Ni}(\text{OH})_2$. (b) Reaction kinetic mechanism diagram.

edge of $\text{Ni}(\text{OH})_2$ surface, initial proton adsorption and final H_2 desorption would be easily occurred in HER. After Ov formation, it can be seen that the Pt1 site replaced the Pt2 site as the most active site. Meantime, the ΔG_{H^*} value of Pt1 site in Ov-Pt/ $\text{Ni}(\text{OH})_2$ was -0.20 eV , which was closer to zero than Pt2 sites in Pt/ $\text{Ni}(\text{OH})_2$ and suggested Ov would greatly promote hydrogen spill-over, and then achieve efficient HER activity. In conclusion, the synergistic effect of Pt-Ov sites can effectively

regulate the localized charge density, and then balance the adsorption and desorption of H^+ . Based on the above research, reaction kinetic mechanism were showed in Fig. 8b. Notably, H_2O preferentially adsorbed on the surface of $\text{Ni}(\text{OH})_2$ and decomposed to produce H^+ , which would gain electrons. Finally, H^+ spilled from $\text{Ni}(\text{OH})_2$ and transfers to Pt surface, and then become H_2 .

4. Conclusion

In summary, we reported a novel MOF derived nanosheets-nanoparticle-structured Pt2/Ni(OH)2/NF, which has abundant O vacancies, SMSI at interface and excellent superhydrophilic surface. Pt2/Ni(OH)2/NF showed excellent HER performance, requiring merely 283 mV at 1000 mA cm⁻², which was even much better than Pt/C catalyst materials. Meanwhile, benefitting from the strong interaction of Pt-O-Ni, the catalyst also showed high stability for 200 h. Assembling our electrodes with a photovoltaic cell yielded up to 13.5% STH conversion efficiency. And the cell voltage for the AEM seawater electrolyzer with the Pt2/Ni(OH)2/NF cathode was 1.46 V at 10 A cm⁻². Corresponding power consumption of Pt2/Ni(OH)2/NF = |NF| was 3.8 kW·h, below that of NF = |NF| (4.3 kW·h) under 200 mA·cm⁻². DFT calculation showed synergistic effect of SMSI of Pt-O-Ni bonds and Ov could effectively regulate the localized charge density and electronic structure. Pt nanoparticles and Ov could balance H^{*} adsorption and desorption and optimize hydrogen spillover from Ni(OH)₂ to Pt sites.

CRediT authorship contribution statement

Jianpeng Sun: Conceptualization, Methodology, Investigation, Writing – original draft preparation, Writing – review & editing. **Zisheng Zhang:** Methodology, Investigation, Writing – review & editing. **Xiangchao Meng:** Supervision, Conceptualization, Writing – original draft, Writing – review & editing.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

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Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at [doi:10.1016/j.apcatb.2023.122703](https://doi.org/10.1016/j.apcatb.2023.122703).

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